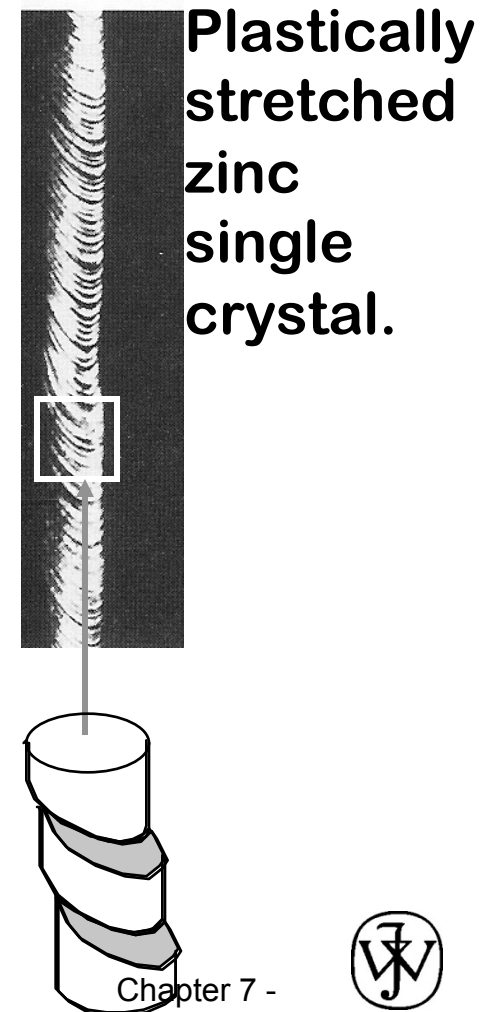


Chapter 7: Dislocations and strengthening mechanisms

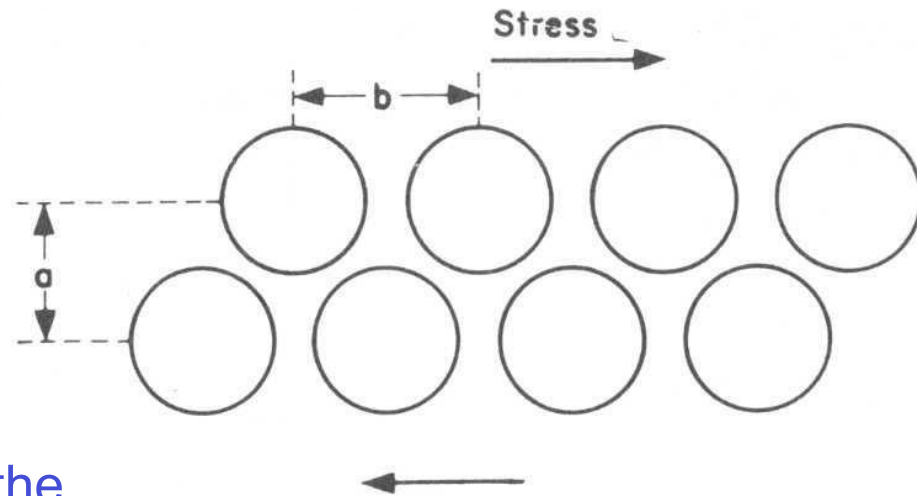
- Introduction
- Basic concepts
- Characteristics of dislocations
- Slip systems
- Slip in single crystals
- Plastic deformation of polycrystalline materials



Theoretical stress

Theoretical stress (Frenkel in 1926)

$$\tau = \frac{Gb}{2\pi a} \sin \frac{2\pi x}{b}$$



G: shear modulus

b: spacing between atoms in the direction of shear stress

a: spacing of the rows of atoms

x: shear translation



Theoretical stress (*continued*)

Hook's law

assumption: small strain

$$\tau = \frac{Gb}{2\pi a} \times \frac{2\pi x}{b} = G \frac{x}{a} = G\gamma$$

Theoretical critical shear stress (maximum stress):

$$\tau_{th} = \frac{b}{a} \frac{G}{2\pi}$$



Theoretical & experimental strength

There is much difference between theoretical and experimental strength

Reasons are:

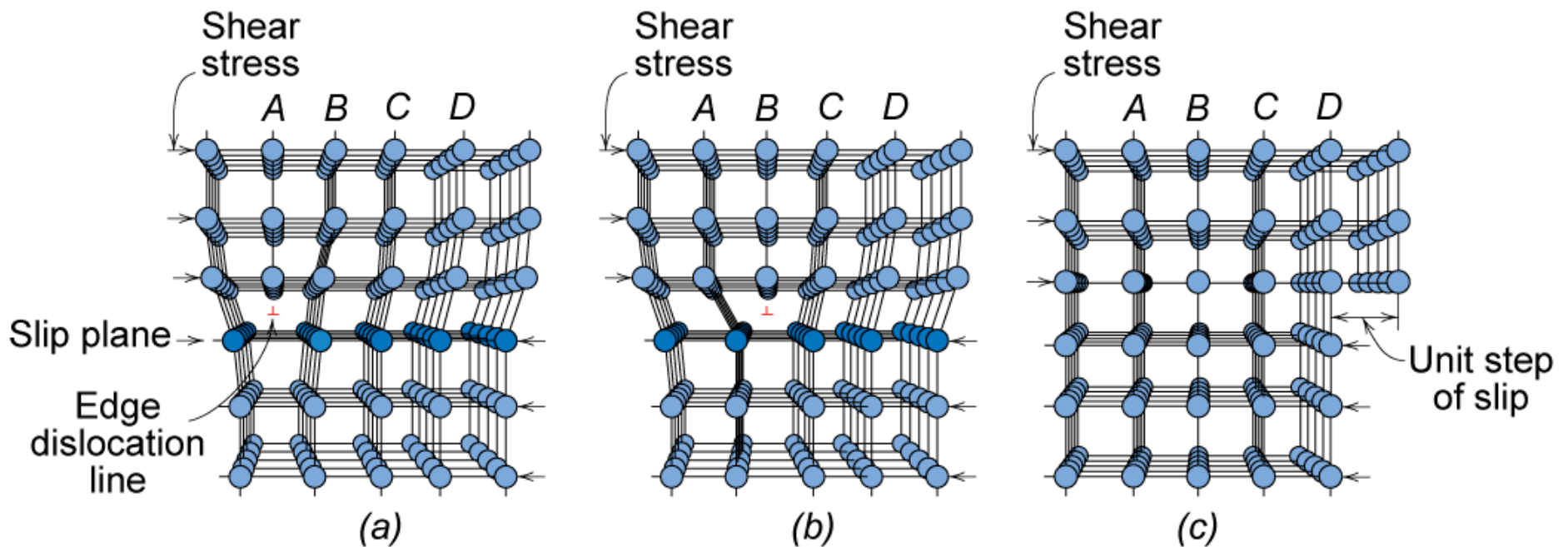
- **Defects are present in all perfect crystal**
- **Dislocation movement makes plastic deformation easier than that predicted by the Frenkel calculation**



Dislocation Motion

Dislocations & plastic deformation

- Cubic & hexagonal metals - plastic deformation by **plastic shear or slip** where one plane of atoms slides over adjacent plane by defect motion (dislocations).



- If dislocations don't move, deformation doesn't occur!

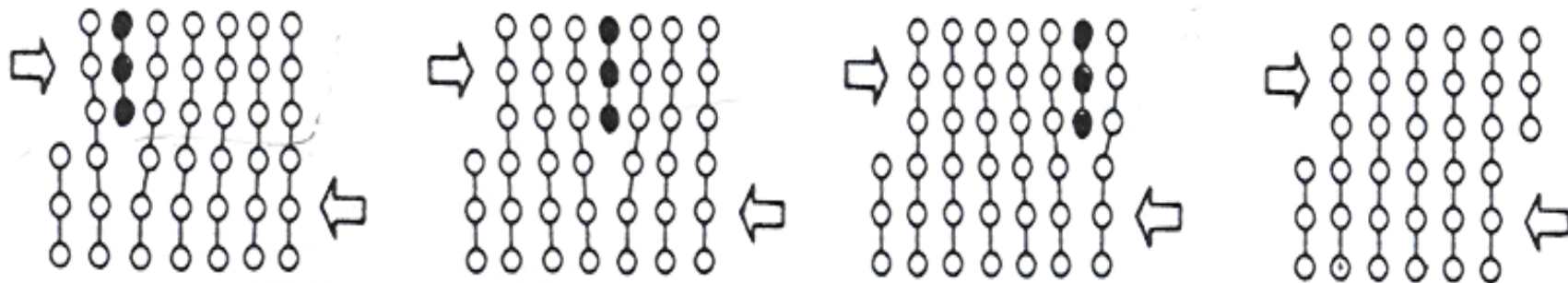
Adapted from Fig. 7.1,
Callister 7e.

Chapter 7 - 5



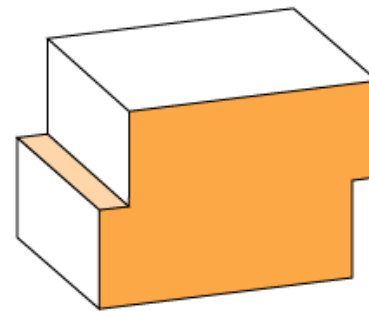
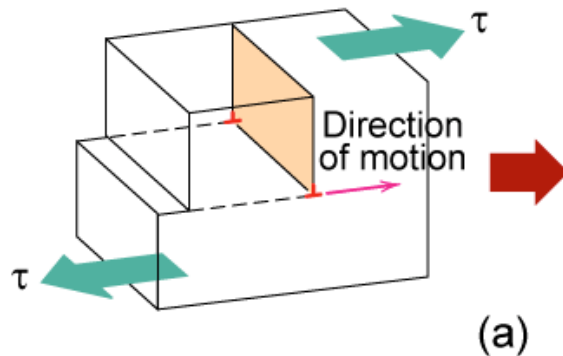
Analogy between caterpillar and dislocation motion

- Dislocation density: total dislocation length per unit volume
- 10^3 mm^{-2} for pure metal crystals; $10^9\text{-}10^{10}\text{mm}^{-2}$ for heavily deformed metals; $10^5\text{-}10^6\text{mm}^{-2}$ for heat-treated deformed metals



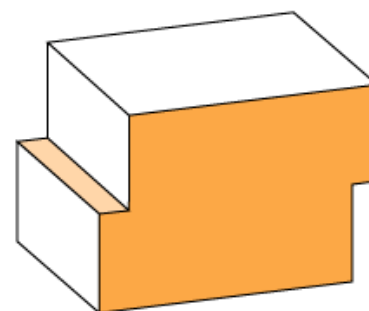
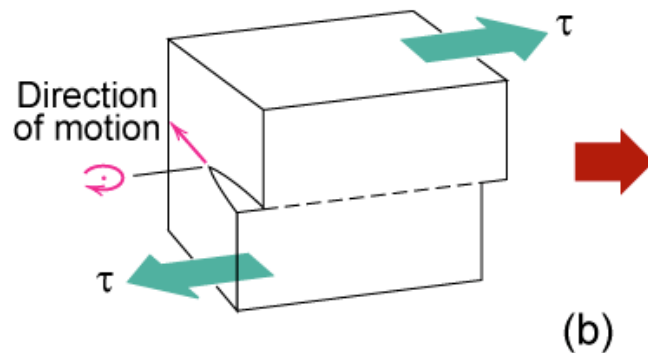
Dislocation Motion

- Dislocation moves along **slip plane** in **slip direction** perpendicular to dislocation line
- Slip direction same direction as **Burgers vector**



Edge dislocation

Adapted from Fig. 7.2,
Callister 7e.



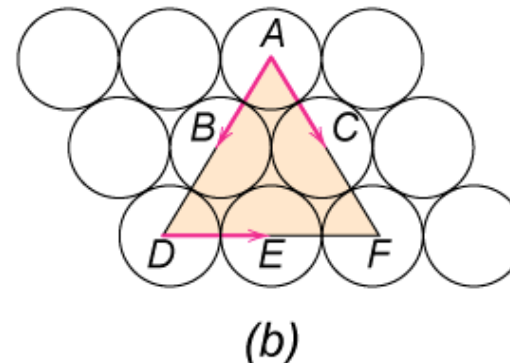
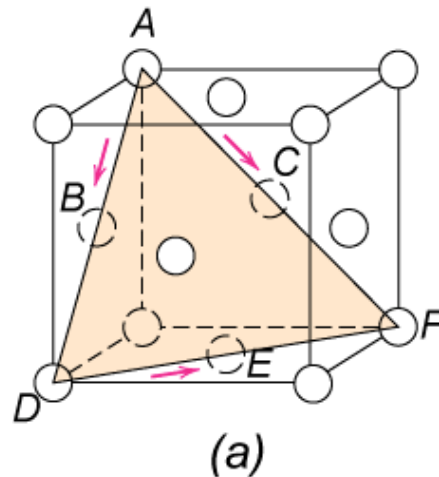
Screw dislocation



Deformation Mechanisms

Slip System

- Slip plane - plane allowing easiest slippage
 - Wide interplanar spacings - highest planar densities
- Slip direction - direction of movement - Highest linear densities



Adapted from Fig. 7.6, Callister 7e.

- FCC Slip occurs on $\{111\}$ planes (close-packed) in $\langle 110 \rangle$ directions (close-packed)
 - => total of 12 slip systems in FCC
- in BCC & HCP other slip systems occur



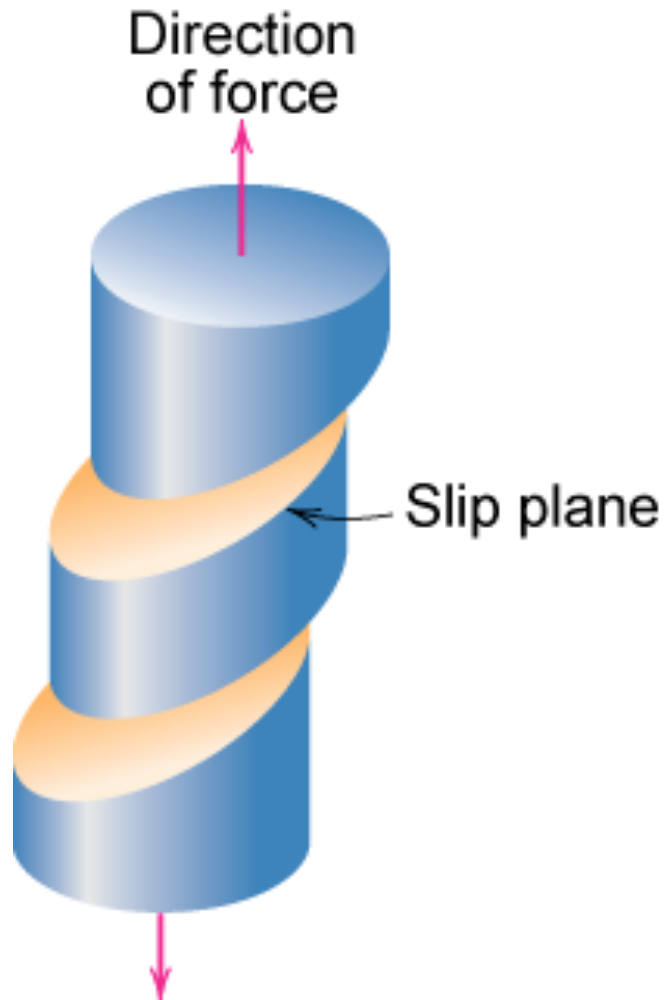
Slip planes and directions for common crystal structure

Table 7.1 Slip Systems for Face-Centered Cubic, Body-Centered Cubic, and Hexagonal Close-Packed Metals

<i>Metals</i>	<i>Slip Plane</i>	<i>Slip Direction</i>	<i>Number of Slip Systems</i>
Face-Centered Cubic			
Cu, Al, Ni, Ag, Au	{111}	$\langle \bar{1}\bar{1}0 \rangle$	12
Body-Centered Cubic			
α -Fe, W, Mo	{110}	$\langle \bar{1}11 \rangle$	12
α -Fe, W	{211}	$\langle \bar{1}11 \rangle$	12
α -Fe, K	{321}	$\langle \bar{1}11 \rangle$	24
Hexagonal Close-Packed			
Cd, Zn, Mg, Ti, Be	{0001}	$\langle 11\bar{2}0 \rangle$	3
Ti, Mg, Zr	{10 $\bar{1}$ 0}	$\langle 11\bar{2}0 \rangle$	3
Ti, Mg	{10 $\bar{1}$ 1}	$\langle 11\bar{2}0 \rangle$	6



Single Crystal Slip



Adapted from Fig. 7.9, *Callister 7e*.

Adapted from Fig. 7.8, *Callister 7e*.



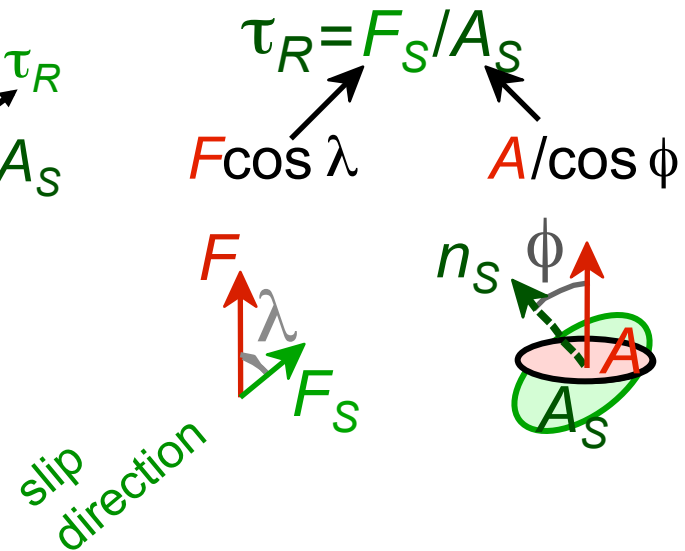
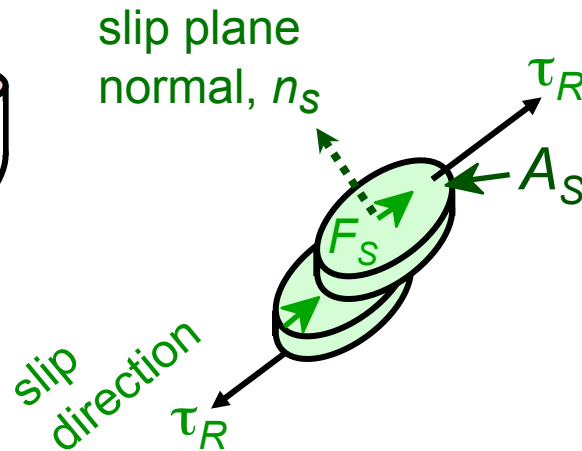
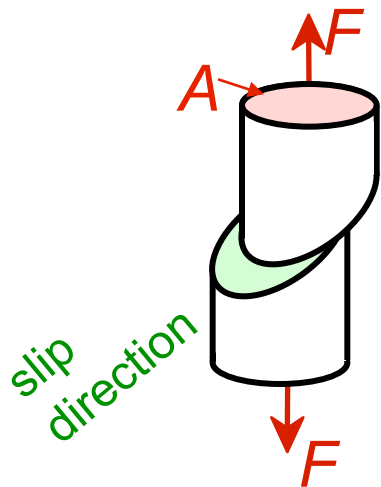
Stress and Dislocation Motion

- Crystals slip due to a **resolved shear stress**, τ_R .
- Applied tension can produce such a stress.

Applied tensile stress: $\sigma = F/A$

Resolved shear stress: $\tau_R = F_S/A_S$

Relation between σ and τ_R



$$\tau_R = \sigma \cos \lambda \cos \phi$$



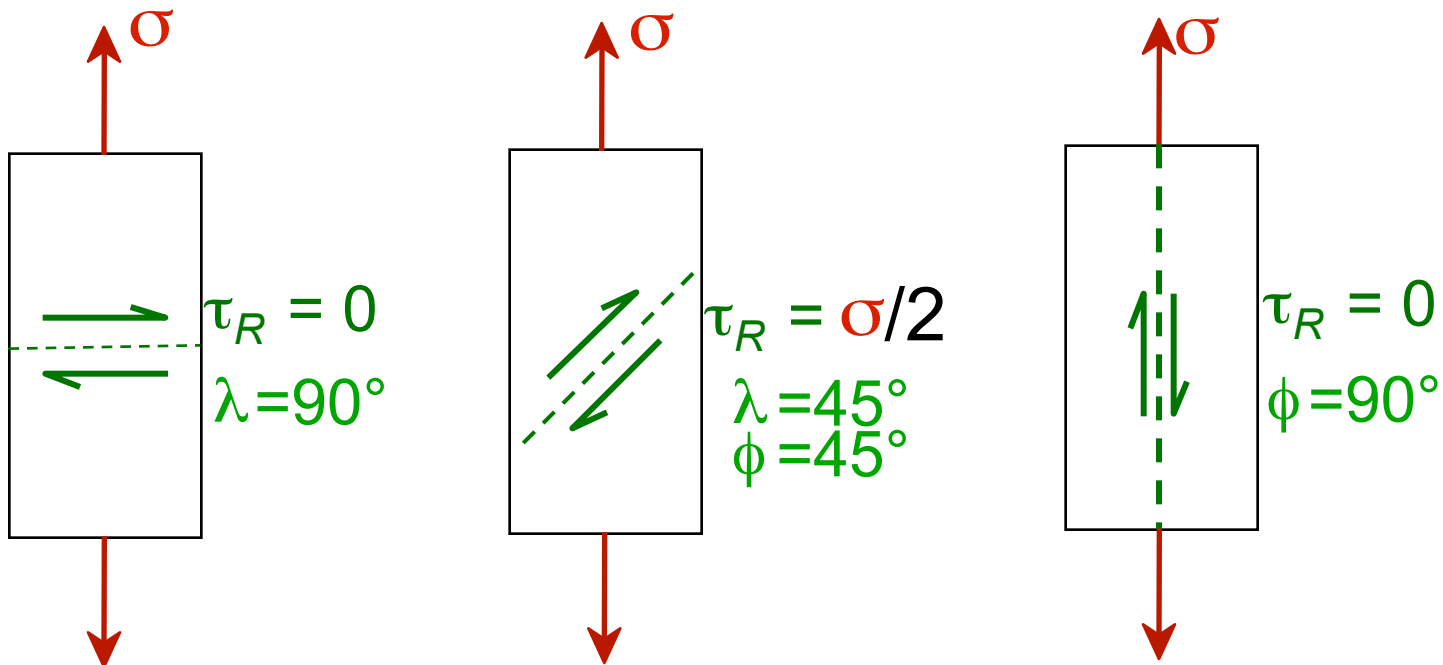
Critical Resolved Shear Stress

- Condition for dislocation motion:
- Crystal orientation can make it easy or hard to move dislocation

$$\tau_R > \tau_{CRSS}$$

↑
typically
 10^{-4} GPa to 10^{-2} GPa

$$\tau_R = \sigma \cos \lambda \cos \phi$$

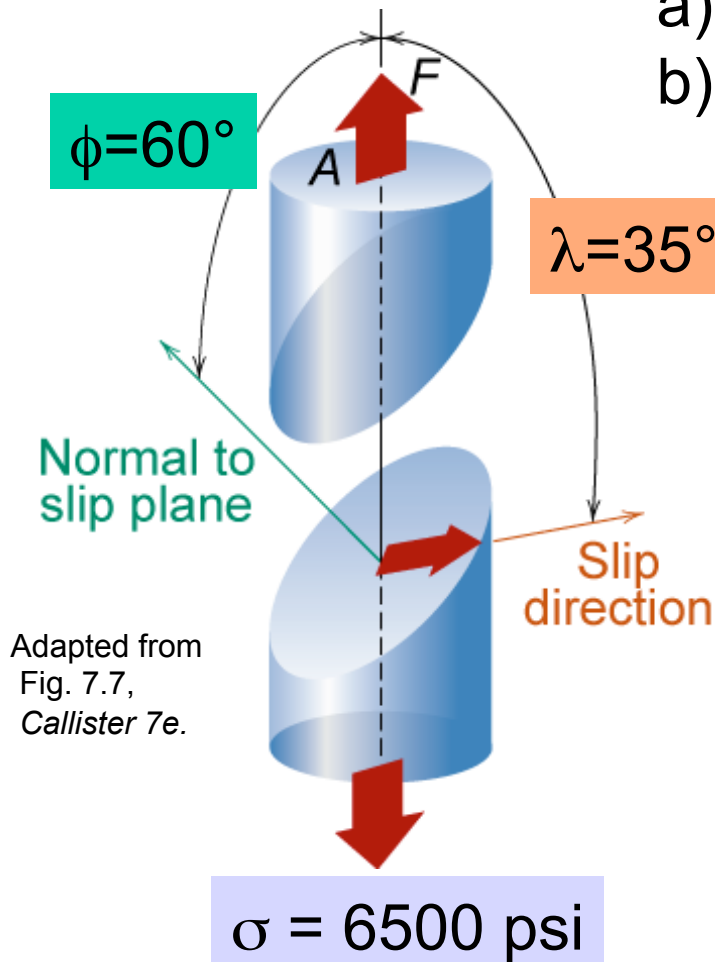


τ maximum at $\lambda = \phi = 45^\circ$



Ex: Deformation of single crystal

- Will the single crystal yield?
- If not, what stress is needed?



$$\tau_{\text{crss}} = 3000 \text{ psi}$$

$$\tau = \sigma \cos \lambda \cos \phi$$

$$\sigma = 6500 \text{ psi}$$

$$\begin{aligned} \tau &= (6500 \text{ psi}) (\cos 35^\circ) (\cos 60^\circ) \\ &= (6500 \text{ psi}) (0.41) \end{aligned}$$

$$\tau = 2662 \text{ psi} < \tau_{\text{crss}} = 3000 \text{ psi}$$

So the applied stress of 6500 psi will not cause the crystal to yield.



Ex: Deformation of single crystal

What stress *is* necessary (i.e., what is the yield stress, σ_y)?

$$\tau_{\text{crss}} = 3000 \text{ psi} = \sigma_y \cos \lambda \cos \phi = \sigma_y (0.41)$$

$$\therefore \sigma_y = \frac{\tau_{\text{crss}}}{\cos \lambda \cos \phi} = \frac{3000 \text{ psi}}{0.41} = \underline{\underline{7325 \text{ psi}}}$$

So for deformation to occur the applied stress must be greater than or equal to the yield stress

$$\sigma \geq \sigma_y = 7325 \text{ psi}$$

Slip Motion in Polycrystals

- Stronger - grain boundaries pin deformations
- Slip planes & directions (λ , ϕ) change from one crystal to another.
- τ_R will vary from one crystal to another.
- The crystal with the largest τ_R yields first.
- Other (less favorably oriented) crystals yield later.



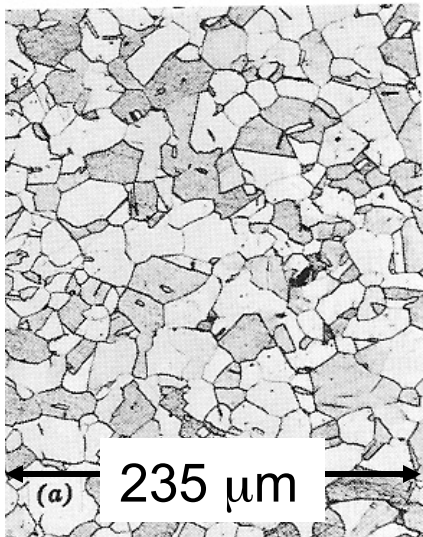
Adapted from Fig. 7.10, *Callister 7e*. (Fig. 7.10 is courtesy of C. Brady, National Bureau of Standards [now the National Institute of Standards and Technology, Gaithersburg, MD].)



Anisotropy in σ_y

- Can be induced by rolling a polycrystalline metal

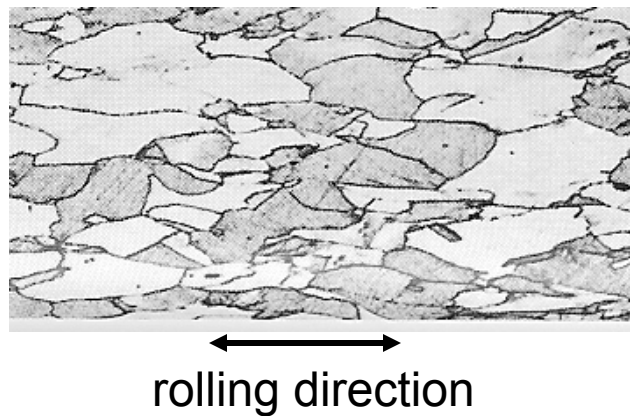
- before rolling



- isotropic

since grains are approx. spherical & randomly oriented.

- after rolling



- anisotropic

since rolling affects grain orientation and shape.

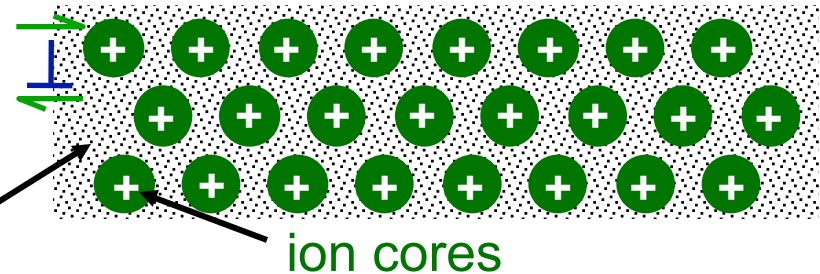
Adapted from Fig. 7.11, *Callister 7e*. (Fig. 7.11 is from W.G. Moffatt, G.W. Pearsall, and J. Wulff, *The Structure and Properties of Materials*, Vol. I, *Structure*, p. 140, John Wiley and Sons, New York, 1964.)



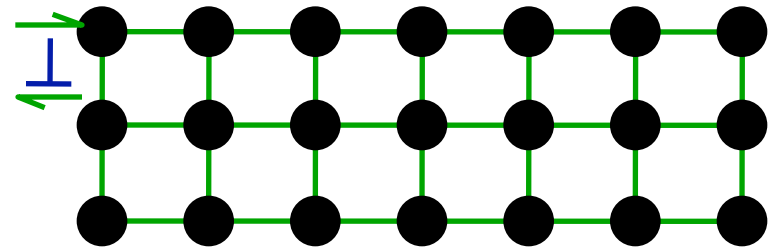
Dislocations & Materials Classes

- Metals: Disl. motion easier.
 - non-directional bonding
 - close-packed directions for slip.

electron cloud



- Covalent Ceramics (Si, diamond): Motion hard.
 - directional (angular) bonding



- Ionic Ceramics (NaCl): Motion hard.
 - need to avoid ++ and -- neighbors.

