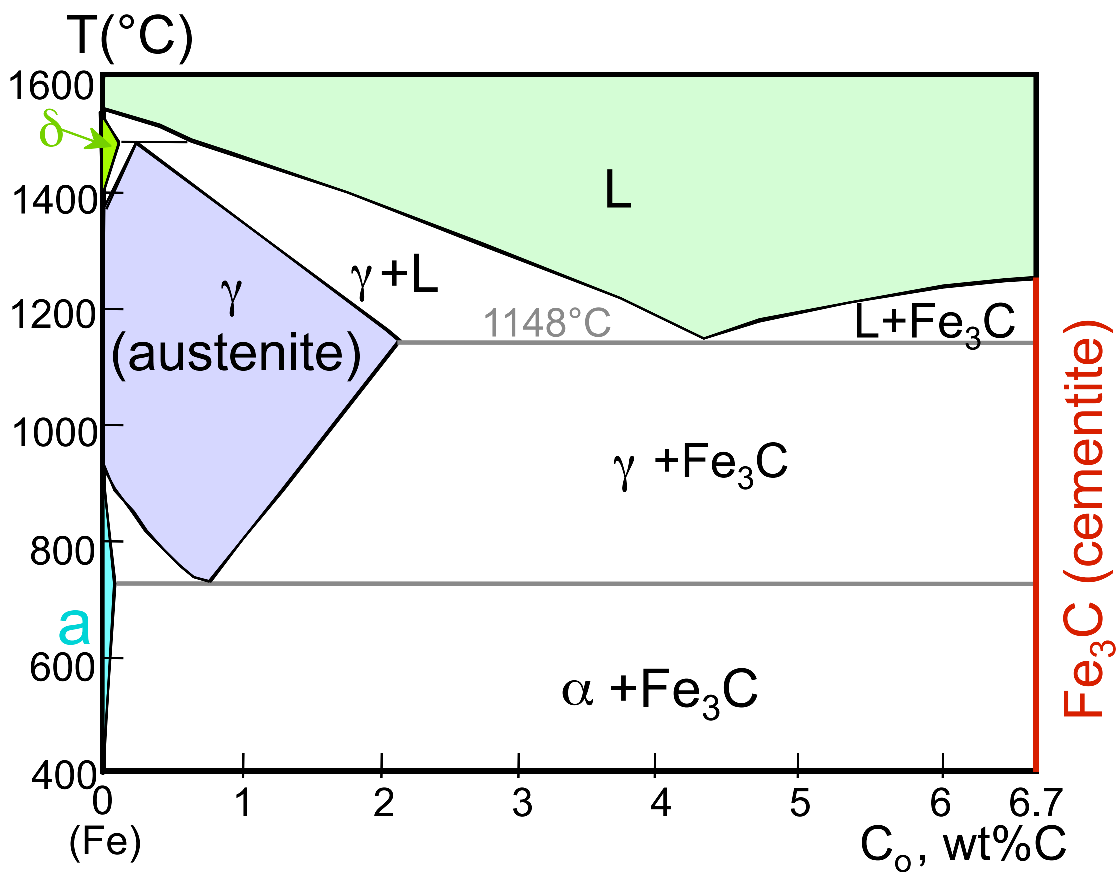
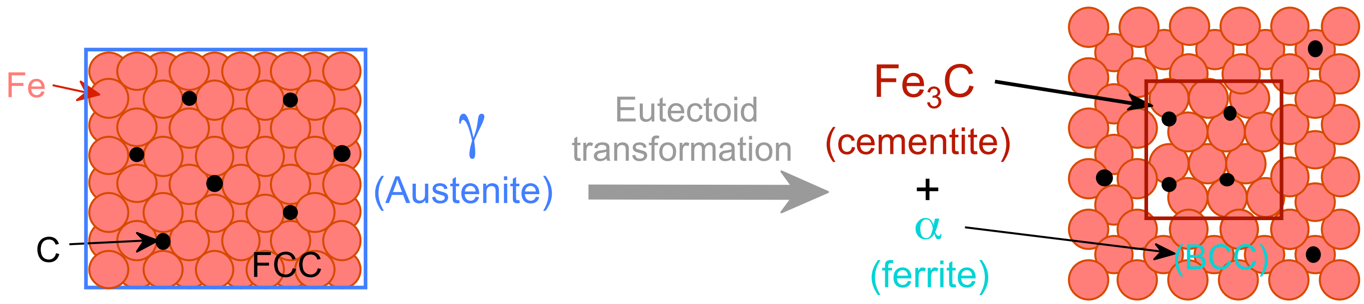
Chapter 10: Phase Transformations



Chapter 10: Phase Transformations

- Transforming one phase into another takes time.



- How does the rate of transformation depend on time and *T*?
- How can we slow down the transformation so that we can engineer non-equilibrium structures?
- Are the mechanical properties of non-equilibrium structures better?

t=time!

T=temperature!

ΔT= temperature difference from phase transition temperature

Phase Transformations

Nucleation

– nuclei (seeds) act as template to grow crystals

Driving force to nucleate increases as we increase ΔT

In general, rate of nucleation higher with higher Δ T

– supercooling (eutectic, eutectoid)

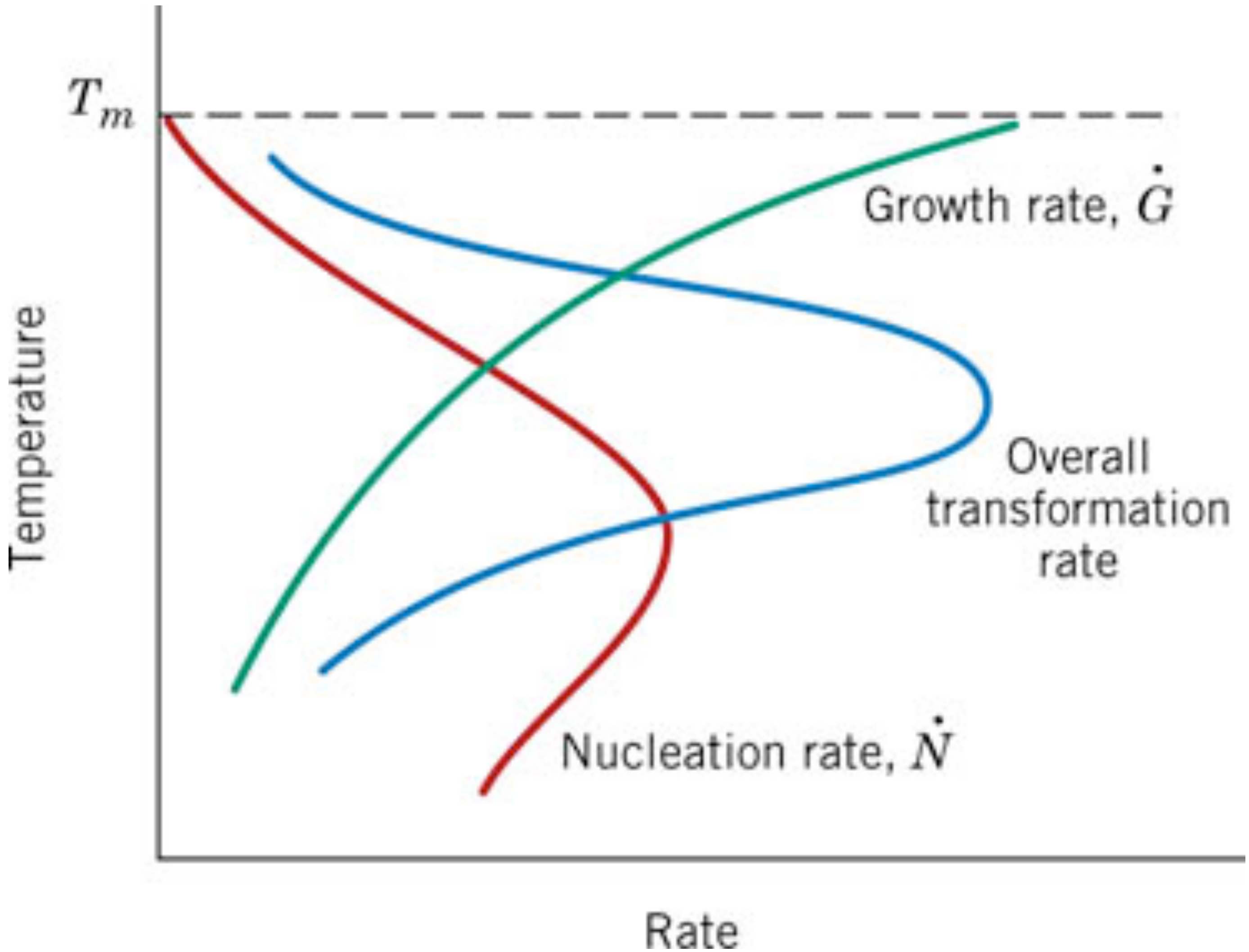
Growth

Growth rate increases with T (thermally activated)

$$dG/dt = C \exp(-Q/kT)$$

Small supercooling → few nuclei - large crystals

Large supercooling → rapid nucleation - many nuclei, small crystals

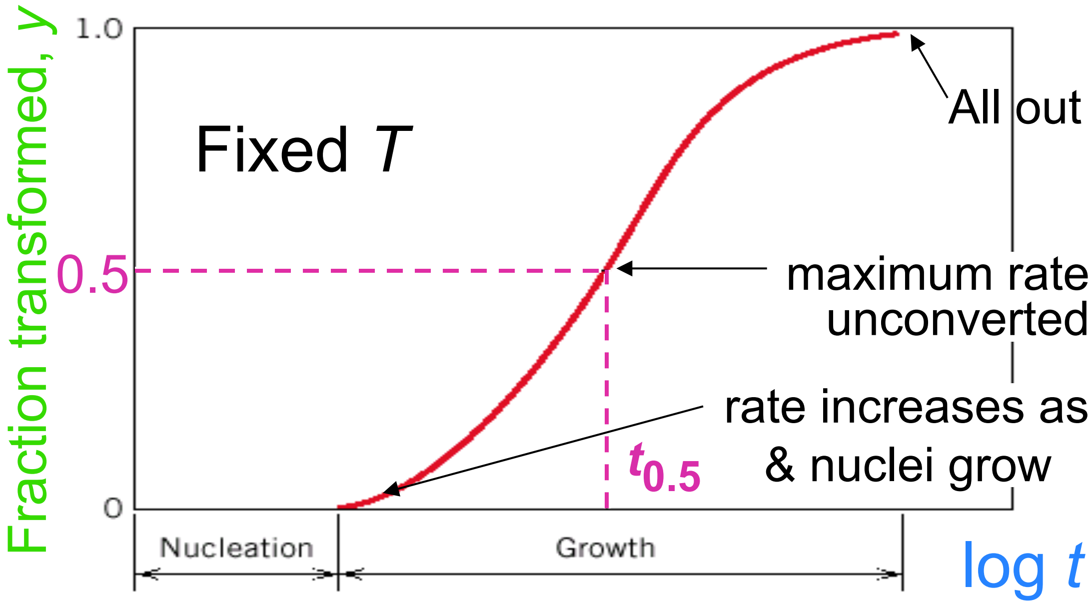

T_m
Growth rate, $\dot{G}$
Temperature
Overall transformation rate
Nucleation rate, $\dot{N}$
Rate

Rate of Phase Transformations

Kinetics - measure approach to equilibrium vs. time

- Hold temperature constant & measure conversion vs. time
  - How is conversion measured?
    - X-ray diffraction – have to do many samples
    - electrical conductivity – follow one sample
    - sound waves – one sample

Rate of Phase Transformation



Adapted from Fig. 10.10, *Callister 7e.*

Avrami rate equation => $y = 1 - \exp(-kt^n)$

fraction transformed — **time**

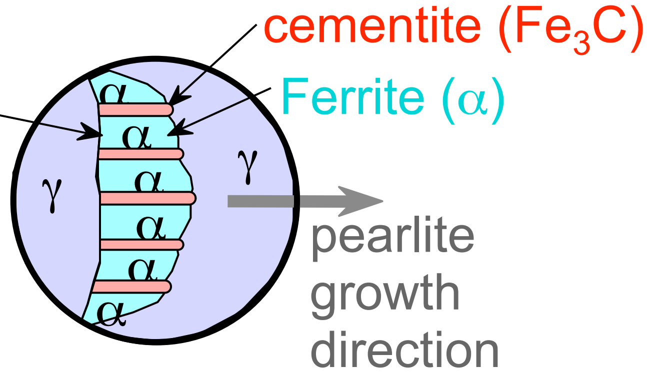
– *k* & *n* fit for specific sample

By convention $r = 1 / t_{0.5}$

Eutectoid Transformation Rate

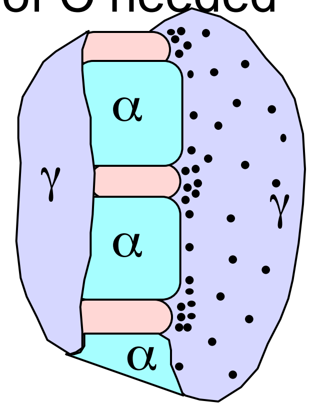
- Growth of pearlite from austenite:

Austenite (γ) grain boundary

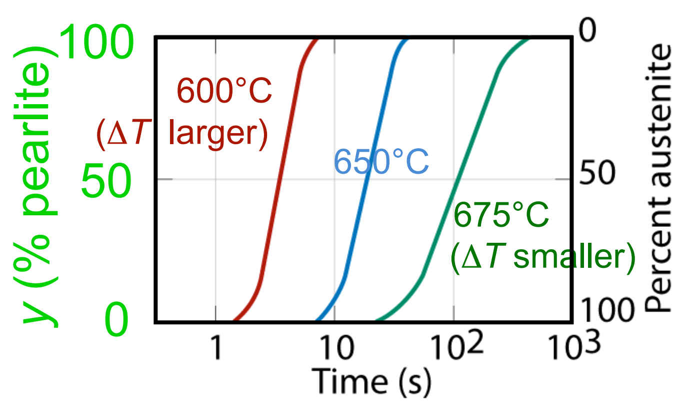


Adapted from Fig. 9.15, *Callister 7e*.

Diffusive flow of C needed



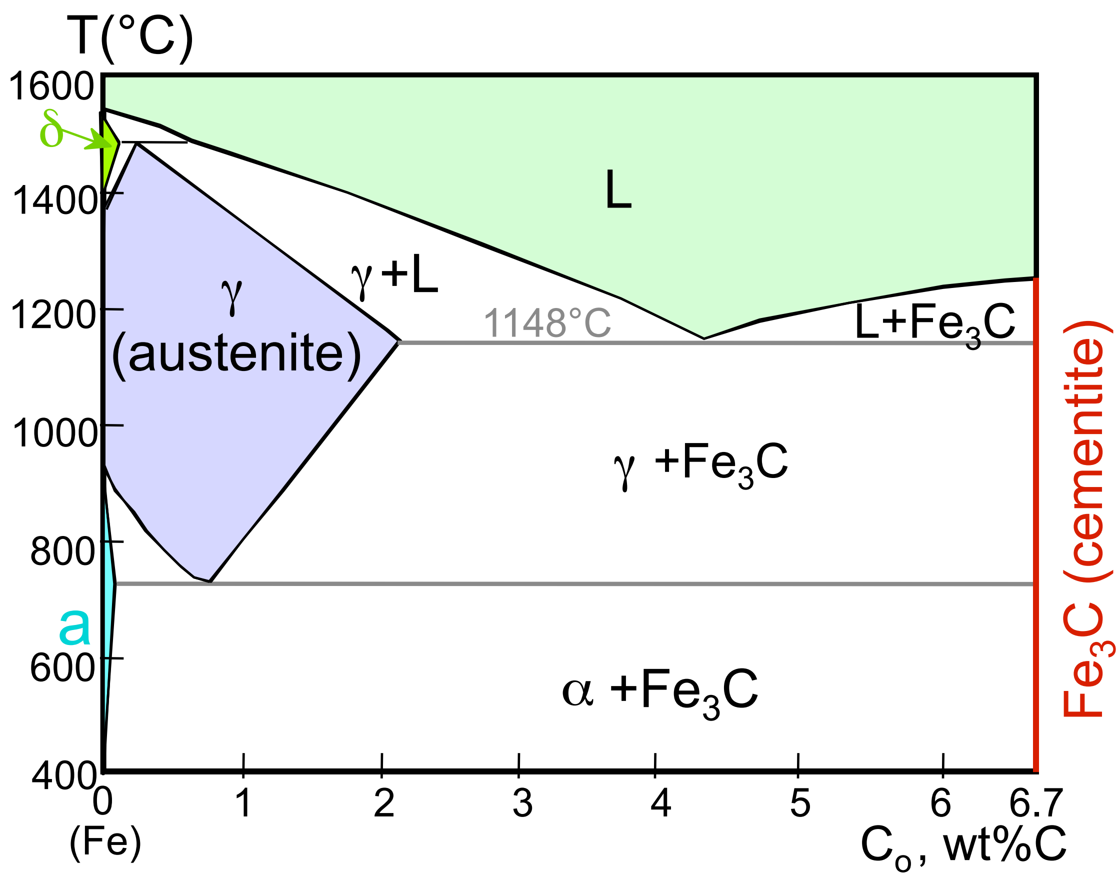
- Recrystallization rate increases with ΔT.



Adapted from Fig. 10.12, *Callister 7e*.

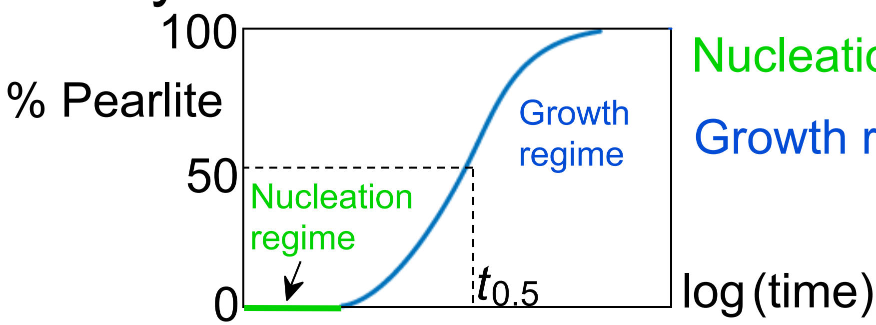
Course pearlite → formed at higher T - softer

Fine pearlite → formed at low T - harder


T(°C)
1600
1400
1200
1000
800
600
400
δ
L
γ+L
γ
(austenite)
1148°C
L+Fe3C
γ +Fe3C
α +Fe3C
a
Fe3C (cementite)
0
(Fe)
1
2
3
4
5
6
6.7
Co, wt%C

Nucleation and Growth

- Reaction rate is a result of nucleation <u>and</u> growth of crystals.

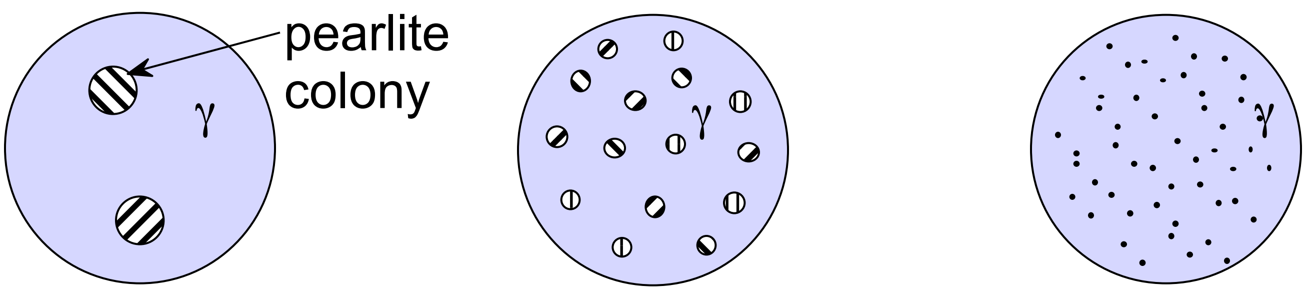


Nucleation rate increases with ΔT

Growth rate increases with T

Adapted from
Fig. 10.10, *Callister 7e*.

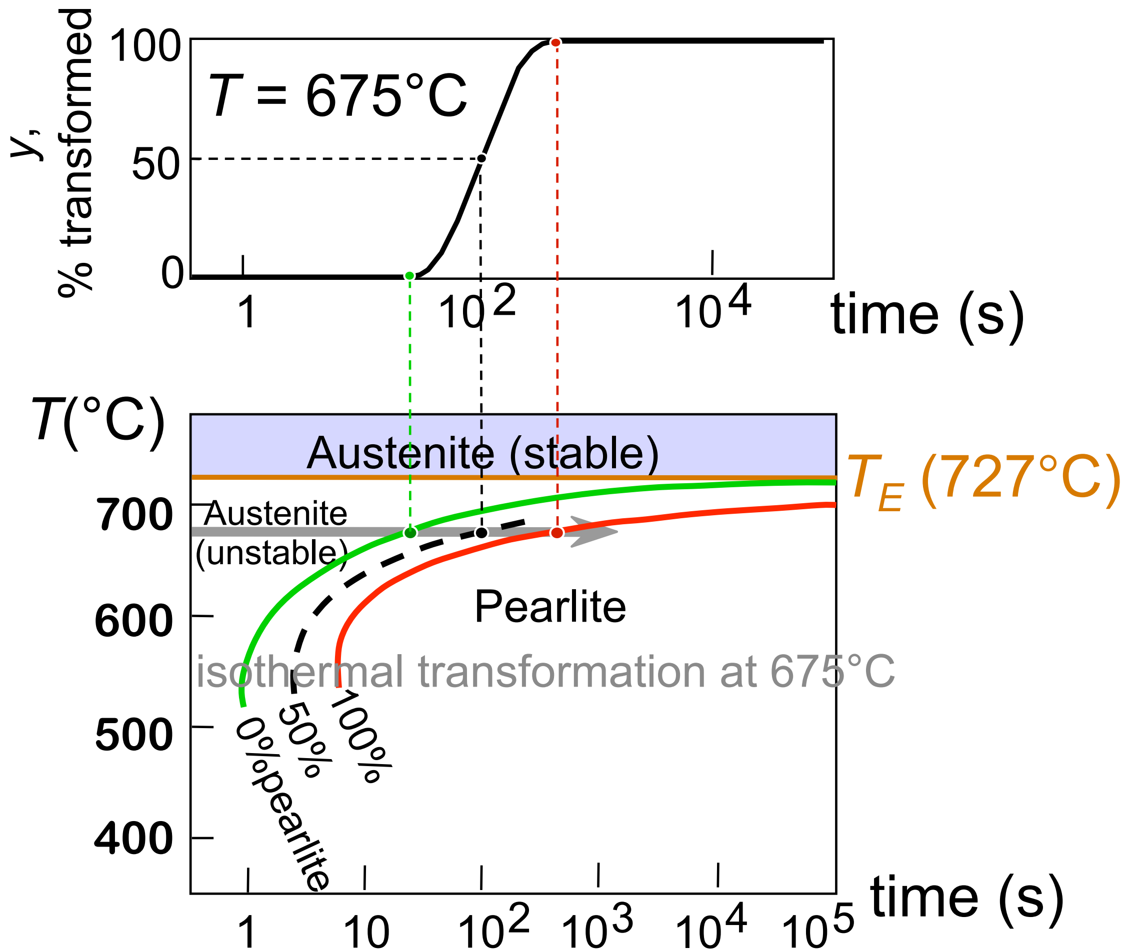
- Examples:



T just below T_E	T moderately below T_E	T way below T_E
Nucleation rate low	Nucleation rate med .	Nucleation rate high
Growth rate high	Growth rate med.	Growth rate low

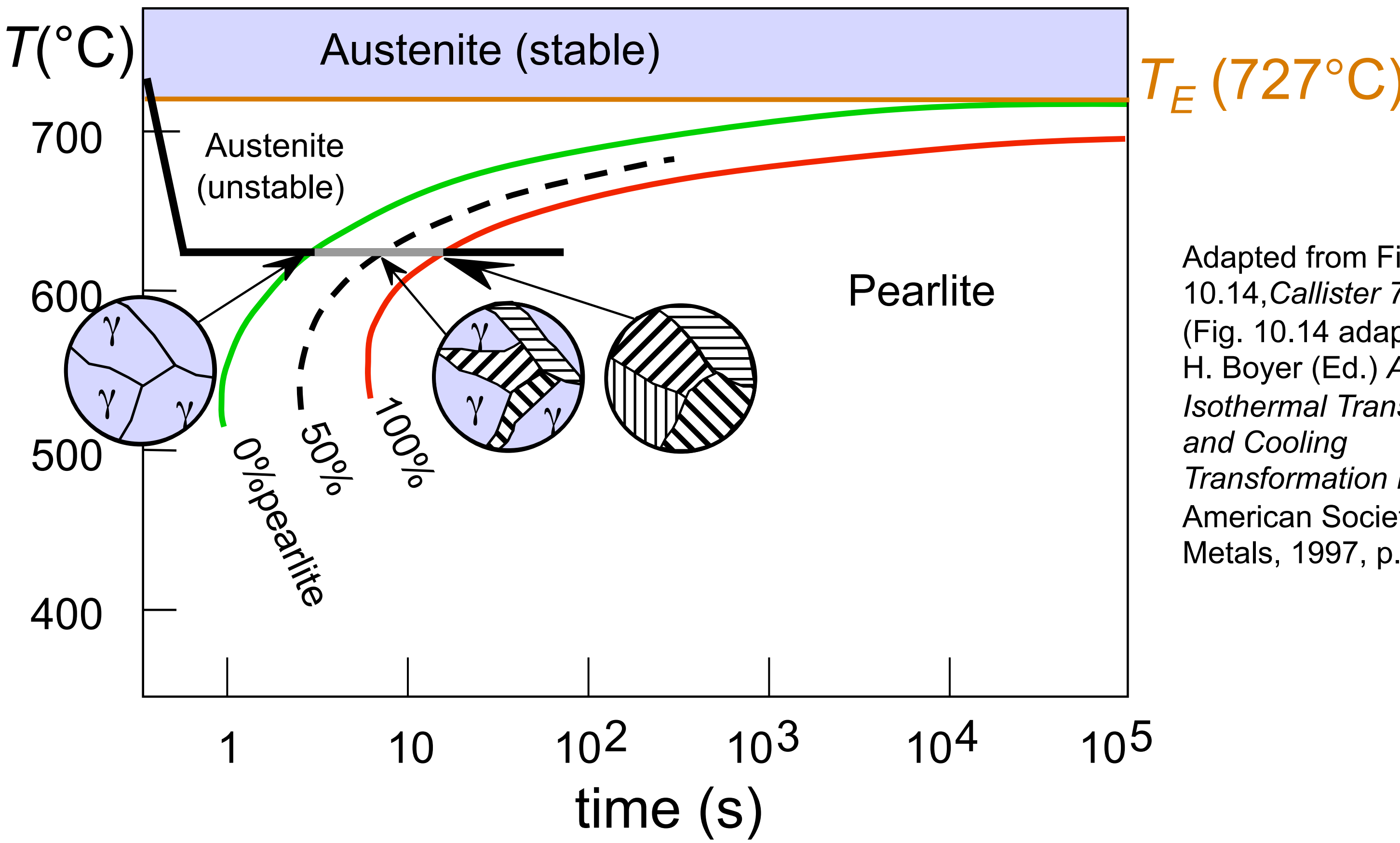
Isothermal Transformation Diagrams

- Fe-C system, C_o = 0.76 wt% C
- Transformation at T = 675°C.



Effect of Cooling History in Fe-C System

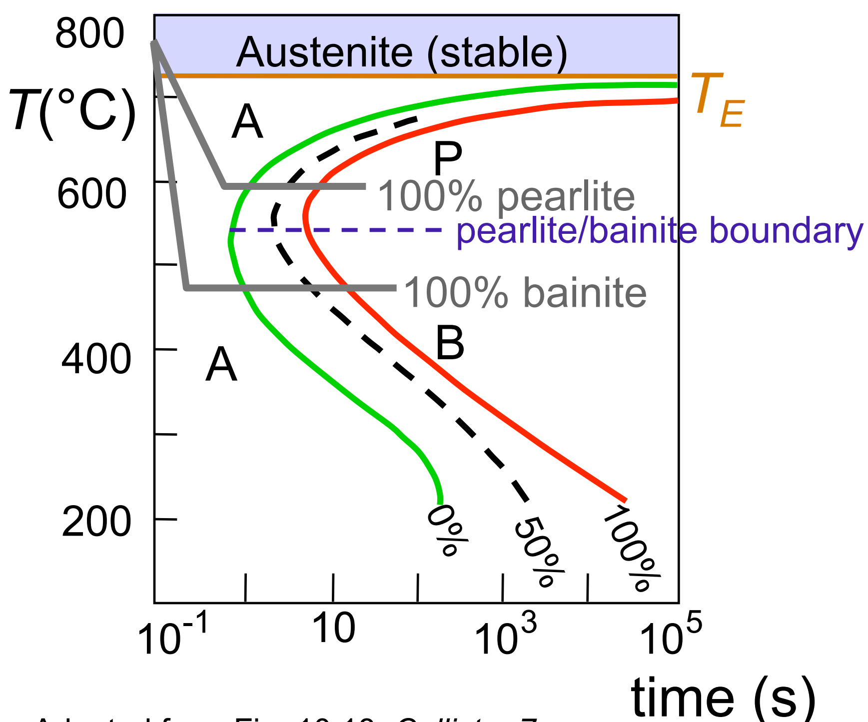
- Eutectoid composition, C_o = 0.76 wt% C
- Begin at T > 727°C
- Rapidly cool to 625°C and hold isothermally.



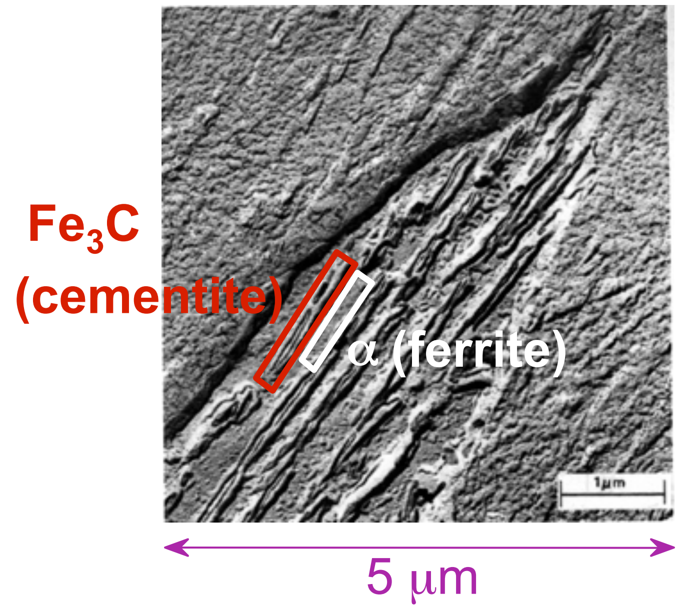
Adapted from Fig. 10.14, *Callister 7e*. (Fig. 10.14 adapted from H. Boyer (Ed.) *Atlas of Isothermal Transformation and Cooling Transformation Diagrams*, American Society for Metals, 1997, p. 28.)

Non-Equilibrium Transformation Products: Fe-C

- Bainite:
  --α lathes (strips) with long rods of Fe_3C
  --diffusion controlled.
- Isothermal Transf. Diagram



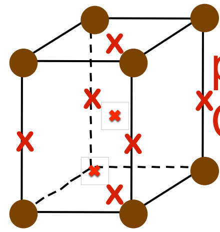
Adapted from Fig. 10.18, *Callister 7e*.
(Fig. 10.18 adapted from H. Boyer (Ed.) *Atlas of Isothermal Transformation and Cooling Transformation Diagrams*, American Society for Metals, 1997, p. 28.)



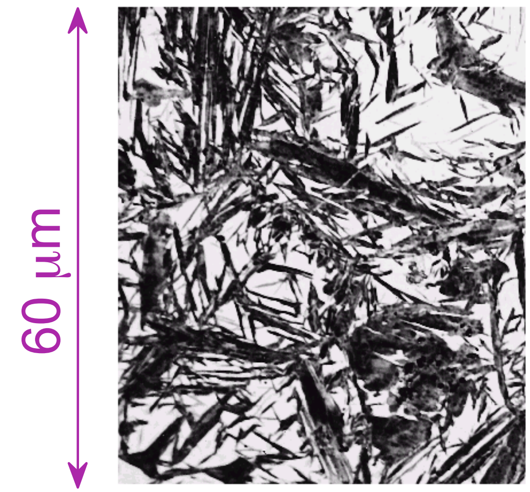
(Adapted from Fig. 10.17, *Callister, 7e*. (Fig. 10.17 from *Metals Handbook*, 8th ed., Vol. 8, *Metallography, Structures, and Phase Diagrams*, American Society for Metals, Materials Park, OH, 1973.)

Martensite: Fe-C System

- Martensite:
  --γ(FCC) to Martensite (BCT)
  (involves single atom jumps)



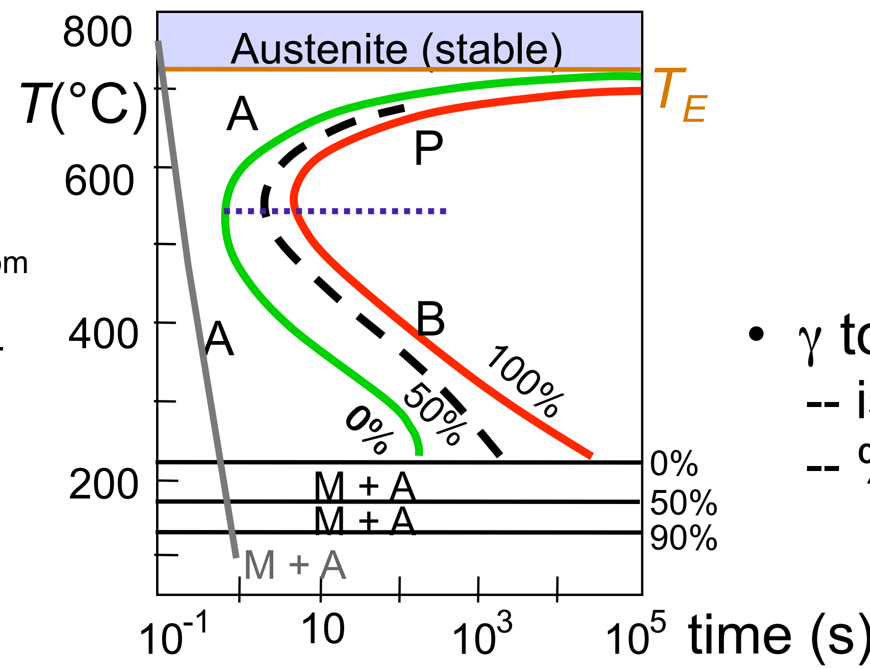
(Adapted from Fig. 10.20, *Callister, 7e*.



Martensite needles
Austenite

(Adapted from Fig. 10.21, *Callister, 7e*.
(Fig. 10.21 courtesy United States Steel Corporation.)

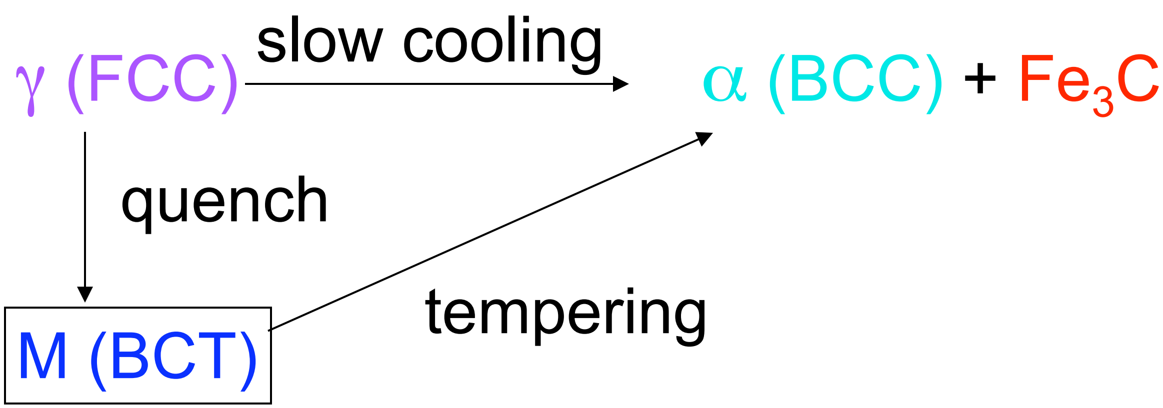
- Isothermal Transf. Diagram



Adapted from Fig. 10.22, *Callister 7e*.

- γ to M transformation..
  -- is rapid!
  -- % transf. depends on T only.

Martensite Formation



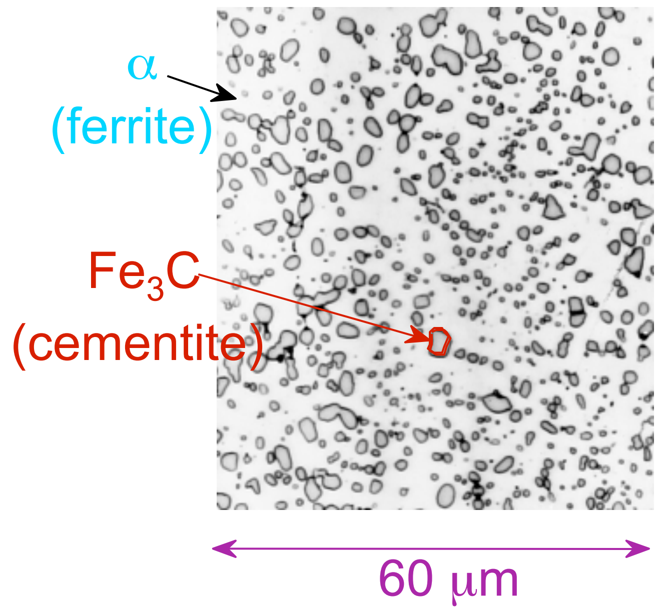
M = martensite is body centered tetragonal (BCT)

Diffusionless transformation BCT if *C* > 0.15 wt%

BCT → few slip planes → hard, brittle

Spheroidite: Fe-C System

- Spheroidite:
  --α grains with spherical Fe_3C
  --diffusion dependent.
  --heat bainite or pearlite for long times
  --reduces interfacial area (driving force)

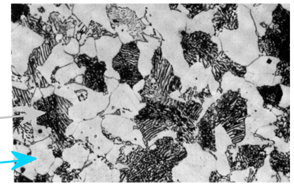


(Adapted from Fig. 10.19, *Callister, 7e*.
(Fig. 10.19 copyright United States Steel Corporation, 1971.)

Mechanical Prop: Fe-C System (1)

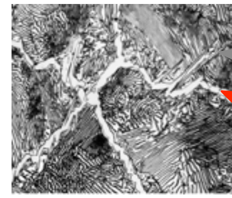
- Effect of wt% C

Pearlite (med)
ferrite (soft)



C_o < 0.76 wt% C
Hypoeutectoid

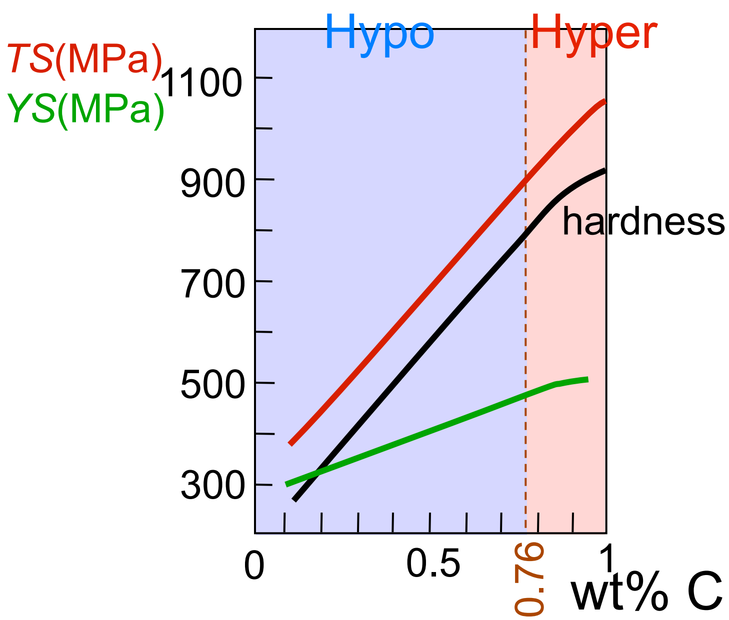
Adapted from Fig. 9.30, *Callister 7e*. (Fig. 9.30 courtesy Republic Steel Corporation.)

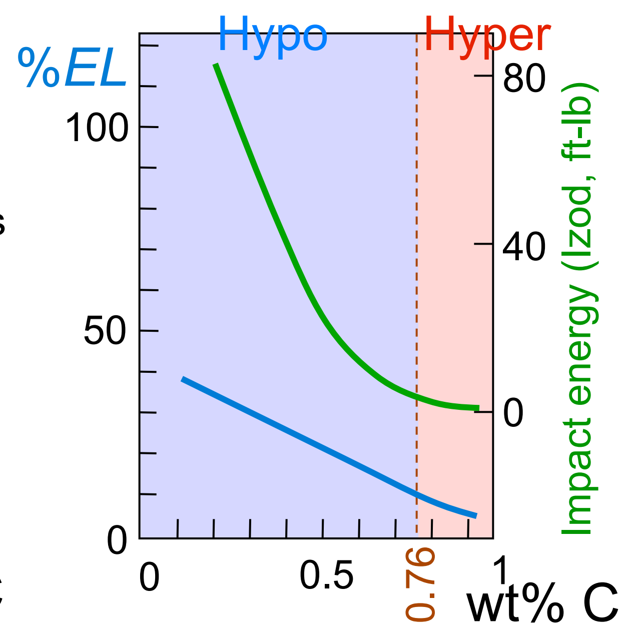


Pearlite (med)
Cementite (hard)

C_o > 0.76 wt% C
Hypereutectoid

Adapted from Fig. 9.33, *Callister 7e*. (Fig. 9.33 copyright 1971 by United States Steel Corporation.)



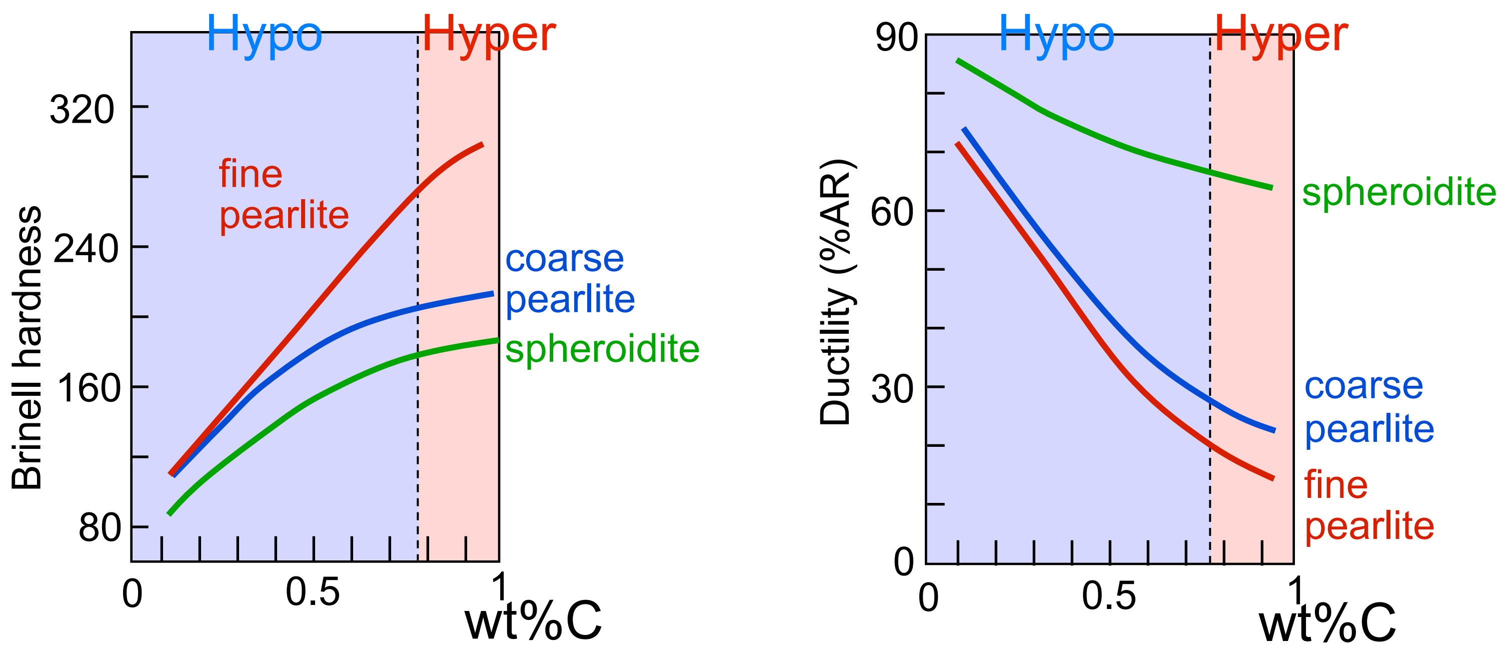


Adapted from Fig. 10.29, *Callister 7e*. (Fig. 10.29 based on data from *Metals Handbook: Heat Treating*, Vol. 4, 9th ed., V. Masseria (Managing Ed.), American Society for Metals, 1981, p. 9.)

- More wt% C: *TS* and *YS* increase, %*EL* decreases.

Mechanical Prop: Fe-C System (2)

- Fine vs coarse pearlite vs spheroidite

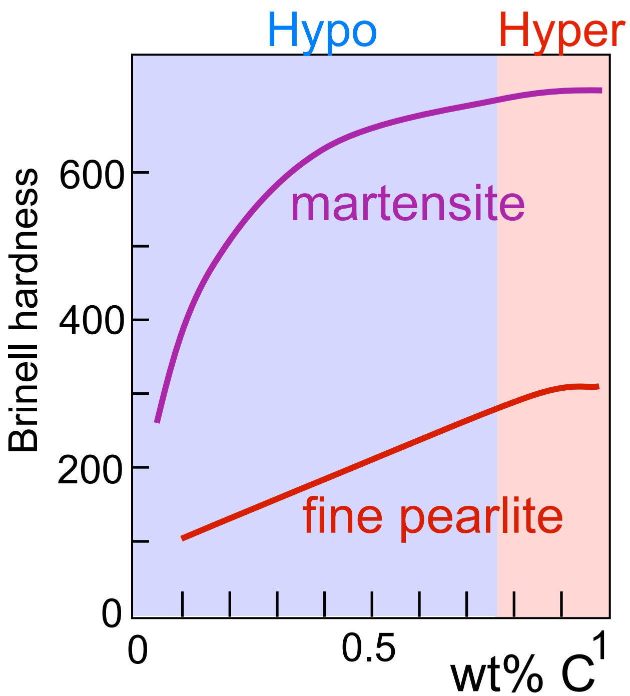


Adapted from Fig. 10.30, *Callister 7e*. (Fig. 10.30 based on data from *Metals Handbook: Heat Treating*, Vol. 4, 9th ed., V. Masseria (Managing Ed.), American Society for Metals, 1981, pp. 9 and 17.)

- Hardness: fine > coarse > spheroidite
- %RA: fine < coarse < spheroidite

Mechanical Prop: Fe-C System (3)

- Fine Pearlite vs Martensite:

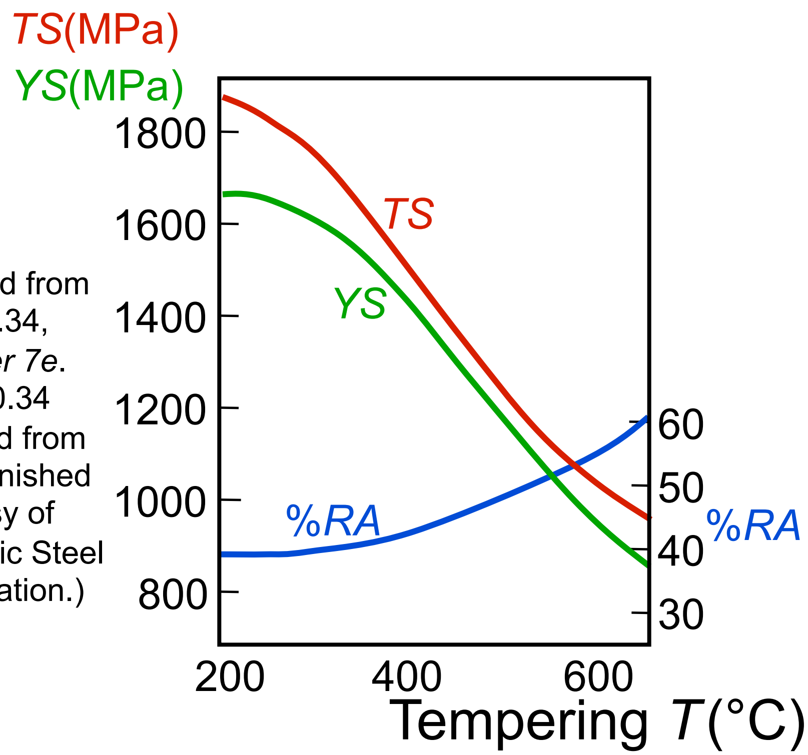


Adapted from Fig. 10.32, *Callister 7e*. (Fig. 10.32 adapted from Edgar C. Bain, *Functions of the Alloying Elements in Steel*, American Society for Metals, 1939, p. 36; and R.A. Grange, C.R. Hribal, and L.F. Porter, *Metall. Trans. A*, Vol. 8A, p. 1776.)

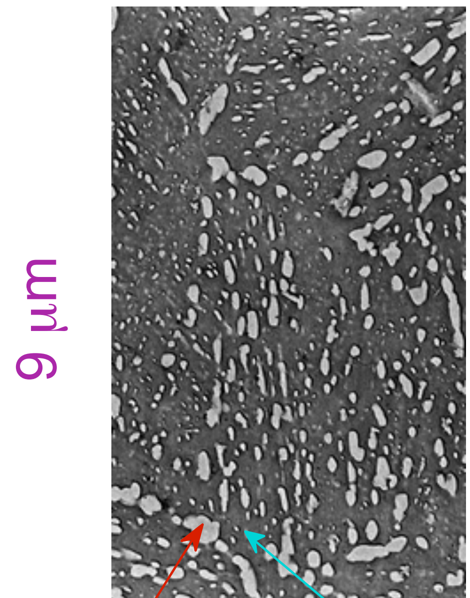
- Hardness: fine pearlite << martensite.

Tempering Martensite

- reduces brittleness of martensite,
- reduces internal stress caused by quenching.



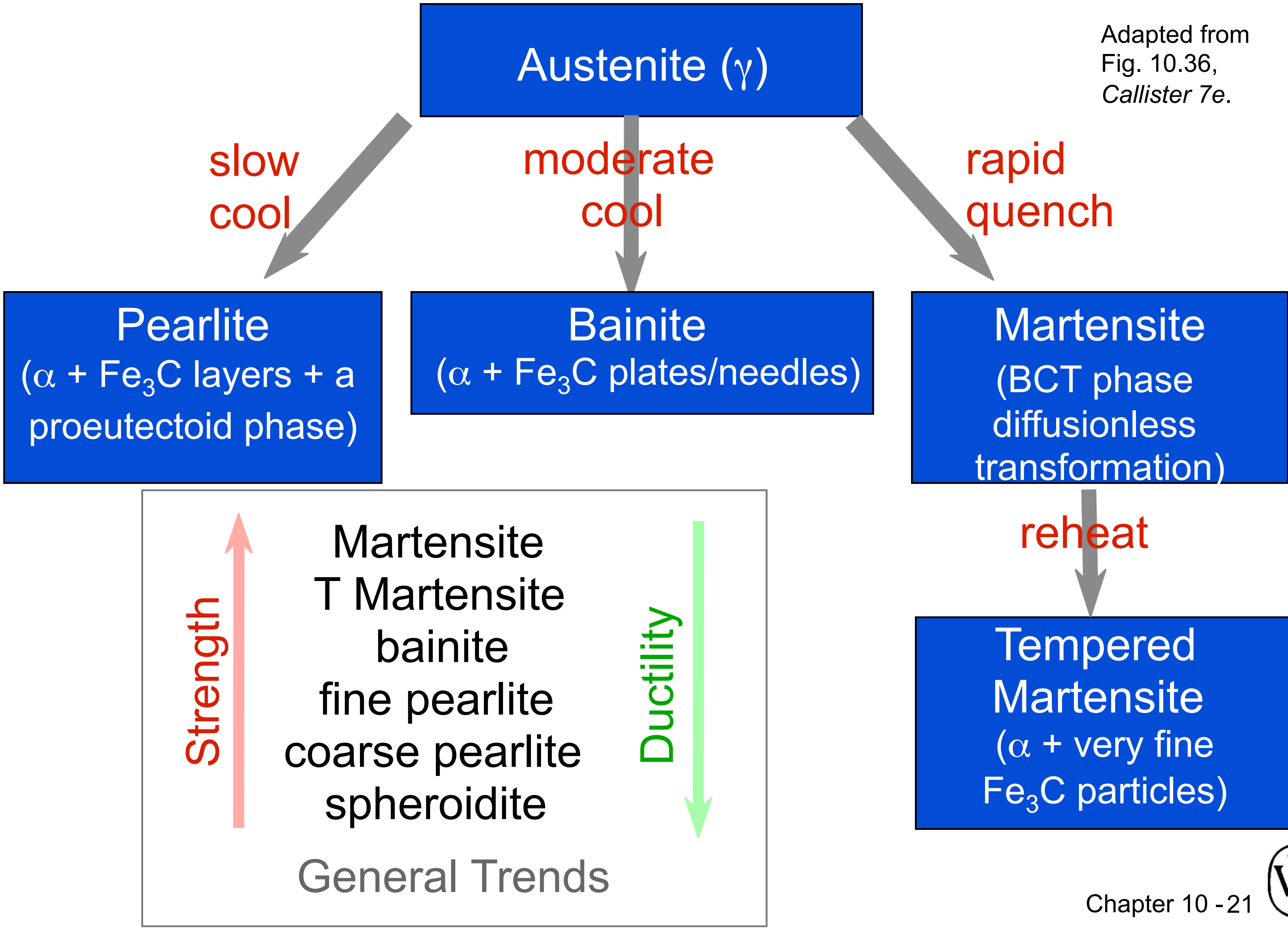
Adapted from Fig. 10.34, *Callister 7e*. (Fig. 10.34 adapted from Fig. furnished courtesy of Republic Steel Corporation.)



Adapted from Fig. 10.33, *Callister 7e*. (Fig. 10.33 copyright by United States Steel Corporation, 1971.)

- produces extremely small Fe_3C particles surrounded by α.
- decreases *TS*, *YS* but increases %*RA*

Summary: Processing Options



Adapted from Fig. 10.36, *Callister 7e*.